

Electrical properties of $\text{AgY}_{1-x}\text{Nd}_x(\text{WO}_4)_2$ solid solutions

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Molybdates and tungstates doped with rare-earth ions (RE^{3+}) are an important group of inorganic materials that have great potential application in various fields. They are extensively studied and more often applied in such luminescent devices as diode-pumped solid-state lasers, phosphor-converted white light-emitting diodes, optical fibres, and integrated optics [1-3]. Silver and rare-earth metal tungstates, $\text{AgRE}(\text{WO}_4)_2$ ($\text{RE} = \text{Y}, \text{Ce-Lu}$), show polymorphism and they crystallize in two polymorphs with the monoclinic and the tetragonal symmetries [4]. The structure of low-temperature, monoclinic phases (S.G. $C2/m$) can be represented by $(\text{W}_4\text{O}_{18})^{8-}$ polyanions, and REO_8 as well as AgO_8 polyhedra. Two neighbouring REO_8 polyhedra are connected by sharing edges along the b axis, forming isolated RE_2O_{14} units [4]. Low-temperature modifications of $\text{AgY}_{1-x}\text{Nd}_x(\text{WO}_4)_2$ solid solutions ($x = 0.005-0.20$) have been successfully prepared by an annealing of stoichiometric mixtures of following tungstates: Ag_2WO_4 , $\text{Nd}_2(\text{WO}_4)_3$, and $\text{Y}_2(\text{WO}_4)_3$. The mixtures of three tungstates were sintered in air at temperatures ranging from 873 K to 1073 K. The sintered ceramics were examined by XRD and SEM methods.

The electrical resistivity $\rho(T)$ of $\text{AgY}_{1-x}\text{Nd}_x(\text{WO}_4)_2$ has been measured in the 76-400 K temperature range with the aid of the four-probe DC method using a KEITHLEY 6517B Electrometer/High Resistance Meter. The thermoelectric power $S(T)$ was measured in the 300-600 K temperature range with the aid of a Seebeck Effect Measurement System (MMR Technologies, Inc., USA). The $\rho(T)$ and $S(T)$ measurements showed the insulating properties and n -type conduction. At high temperatures, *i.e.* above 400 K, we find a rather well defined linear slope of $S(T) = aT$ (diffusion thermopower) which extrapolates to (0, 0) [5]. The diffusion thermopower for an electron gas is due to carrier diffusion and it contains information about the Fermi edge [6]. The smaller value of the diffusion coefficient a is observed when Nd^{3+} content in $\text{AgY}_{1-x}\text{Nd}_x(\text{WO}_4)_2$ is larger. For the same samples, a reversal of thermoelectric power sign is also being observed. The residual electrical conduction of the type n or p in the solid solutions under study seem to be connected with the anionic or cationic vacancies, respectively, the same as it was observed in the $\text{CdRE}_2\text{W}_2\text{O}_{10}$ compounds [7]. The changes of sign observed in $S(T)$ can be an effect of the different values of the activation energy of the vacancy acceptor and donor levels as well as a transfer of the phonon momentum to the electron gas.

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[1] G. Boulon, *Opt. Mater.* **34**, 499 (2012).

[2] G. Métrat, N. Muhlstein, A. Brenier, and G. Boulon, *Opt. Mater.* **8**, 75 (1997).

[3] A. S. Grabtchikov, A. N. Kuzmin, V. A. Lisinetskii, A. A. Orlovich, A. A. Demidovich, K. Yumashev, N. V. Kuleshov, H. J. Eichler, and M. B. Danailov, *Opt. Mater.* **16**, 349 (2001).

[4] P. V. Klevtsov, and R. F. Klevtsova, *J. Strukt. Chem.* **18**, 339 (1977).

[5] T. Groń, K. Bärner, Ch. Kleeberg, and I. Okońska-Kozłowska, *Physica B* **225**, 191 (1996).

[6] R. D. Barnard, *Thermoelectricity in Metals and Alloys*, Taylor & Francis, London, 1972.

[7] Z. Kukula, E. Tomaszewicz, S. Mazur, T. Groń, S. Pawlus, H. Duda, and T. Mydlarz, *J. Phys. Chem. Solids* **74**, 86 (2013).